Band gap tunability in deformable dielectric composite plates

Roey Getz, Gal Shmuel*

Faculty of Mechanical Engineering, Technion—Israel Institute of Technology, Haifa 32000, Israel

A R T I C L E   I N F O

Article history:
Received 1 May 2017
Revised 6 July 2017
Available online 26 July 2017

Keywords:
Dielectric elastomer film
Wave propagation
Band gap
Composite
Phononic crystal
Plane wave expansion
Nonlinear elasticity

A B S T R A C T

Subject to voltage, dielectric elastomers deform and stiffen. We show that harmonic excitations at certain frequencies cannot propagate in incompressible dielectric elastomer fiber composite films. Importantly, we demonstrate that these band gaps are tunable by the voltage. To show this, we formulate the equations governing small-amplitude waves in a deformed film, taking into consideration its surfaces. We develop a scheme to numerically solve the resultant equations, based on the supercell plane wave expansion method. To arrive at the findings above, we apply our scheme to a composite with circular fibers, and parametrically study the propagation dependency on the phase properties, film thickness, and voltage. Our results are another step towards the use of soft dielectric films as active wave manipulators.

© 2017 Elsevier Ltd. All rights reserved.

1. Introduction

Mechanical waves traveling through elastic composites encounter periodic boundaries, which create secondary waves and, in turn, an interference pattern. This interference prevents the propagation of waves at particular frequency intervals, referred to as Bragg band gaps (Sigalas and Economou, 1992; Kushwaha et al., 1993; Hussein et al., 2014). Therefore, elastic composites exhibiting band gaps can potentially protect from undesired vibrations (Wen et al., 2005; Matlack et al., 2016) and filter noise (Vasseur et al., 2002; Babae et al., 2016; Bortot and Shmuel, 2017). Since in the process described, energy is redirected rather than dissipated, such composites can have additional functionalities including wave steering and energy tunneling (Srivastava, 2016; Zelhofer and Kochmann, 2017; Ganesh et al., 2017).

The band gap structure depends on the physical and geometrical properties of the composite. Therefore, the functionally of passive composites, which were initially studied and whose characteristics are fixed, is fixed too. To achieve adaptive functionalities, active composite, whose characteristics are tunable by external stimuli, were later on investigated. Specifically, we recall active composites which are deformable by deformation and mechanical instabilities (Bertoldi and Boyce, 2008; Shin et al., 2015; Shmuel and Band, 2016; Barnwell et al., 2017); magnetic stimulus (Matar et al., 2012; Bayat and Gordaninejad, 2015); electric loads (Beck et al., 2011; Degraeve et al., 2015; Shmuel and Salomón, 2016; Celii et al., 2017); and thermal tuning (Ruzzene and Baz, 1999; Jim et al., 2009). In this work, we investigate composites made of dielectric elastomer films (Pelrine et al., 2000; Carpi et al., 2010; Cohen and deBotton, 2016), owing to their simple working principle, low cost, fast response, and ability to experience significant geometrical and physical changes.

Specifically, we consider a film made of two dielectric elastomer phases, which is quasi-statically activated by applying voltage to a pair of compliant electrodes coating its upper and lower surfaces. In the process, opposite charges accumulate on the electrodes, and the resultant Coulomb force between these charges squeezes the film along its thickness, while the elastomer phases stiffen. On top of the deformed configuration, we examine the propagation of incremental waves, while accounting for the reflections at the upper and lower surfaces; in terms of the mechanics terminology—we examine plate or Rayleigh–Lamb modes in the actuated film. Since the actuation changes the geometry and the stiffness of the film, it will change the propagation characteristics (Shmuel et al., 2012). Particularly, it is expected that the actuation will change the range of the gaps. Our objective is to show that indeed the gaps are tunable, and to study how the interplay between the voltage and the physical properties and thickness of the film affects this tunability. Thereby, this paper continues a series of works on tunable band gaps in dielectric elastomer composites; from gaps in the (i) thickness modes of laminates (Shmuel and deBotton, 2012), through (ii) bulk anti-plane (Shmuel, 2013) and (iii) bulk in-plane modes of fiber composites (Getz et al., 2017), to Lamb modes in this paper.

The stages of this work are presented in the following order. Firstly, a framework to analyze the dynamics of deformable dielec-
tric composites \(^1\) is reviewed in Section 2 (Toupin, 1963; McMeeking and Landis, 2005; debottoni et al., 2007; Zhao and Suo, 2010; Dorfmann and Ogden, 2010; Pamies, 2014). This theory is specialized in Section 3, to develop the equations governing small-amplitude waves in a composite film of finite thickness, which was electrostatically deformed. The derived coupled differential equations cannot be solved analytically. To obtain numerical solutions, we formulate a variation of the supercell plane wave expansion (SC-PWE) method (Hou and Assouar, 2008; Vasseur et al., 2008), which accounts for the finite deformation, electroelastic coupling and outer boundaries. Utilizing our formulation, in Section 4 we carry out a parametric study of the dependency of the band diagram on the volume fraction and the physical properties of each constituent, thickness of the film, and applied voltage. Specifically, the numerical analysis shows that the film exhibits complete gaps, namely, independent of the propagation direction and the plane of motion, which are indeed tunable by the applied voltage. A summary of our results concludes the paper in Section 5.

2. Dynamics of dielectric elastomer composites

Consider a two-phase deformable dielectric surrounded by vacuum, occupying the volume \(\Omega_0 \subset \mathbb{R}^3\) and bounded by \(\partial \Omega_0\). A continuous and differentiable function \(\chi : \Omega_0 \times \mathbb{R} \to \mathbb{R}^3\) maps material points \(x \in \Omega_0\) at time \(t\) in the interval \(\mathbb{F} \in \mathbb{R}\) to their current position \(X = \chi(x,t)\). In the current configuration, the body occupies the volume \(\Omega\) and bounded by \(\partial \Omega\). Mappings based on the deformation gradient \(\nabla_X X\), denoted by \(F\), connect line, area and volume elements in the neighborhood of \(X\), denoted \(d\mathbf{X}\), \(dA\), and \(dV\), respectively, and their current configuration counterparts. Specifically, these are \(d\mathbf{X} = Fd\mathbf{X}\), \(dn = \frac{F^{-1}}{n}d\mathbf{N}\) and \(dr = Fdr\), where \(j = \det F > 0\), and \(\mathbf{N}\) and \(n\) are unit normals of referential and deformed area elements, respectively.

We denote by \(\mathbf{e}\) and \(\mathbf{d}\) the electric field and electric displacement field in the current configuration, governed by Maxwell's equations

\[
\nabla \cdot \mathbf{d} = 0, \quad \nabla \times \mathbf{e} = 0, \quad (1)
\]

where \(\nabla \cdot (\cdot)\) and \(\nabla \times (\cdot)\) are the divergence and curl operators, respectively, evaluated with respect to \(x\). The form of the first of Eq. (1) accounts for the absence of free body charge in dielectrics. The second of Eq. (1) uses an electrostatic approximation, when the length of the electromagnetic waves is significantly longer than its mechanical counterpart. In vacuum, \(\mathbf{e}\) and \(\mathbf{d}\) are linearly related through the permittivity of free space \(\varepsilon_0 = 8.85 \cdot 10^{-12} \text{ F m}^{-1}\). In the dielectric, they are related through the electric polarization density field \(\mathbf{p}_e\)

\[
\mathbf{p}_e = \mathbf{d} - \varepsilon_0 \mathbf{e}. \quad (2)
\]

The balance of linear momentum in the presence of electric field is

\[
\nabla \cdot \sigma = \rho \chi_{ttt}, \quad (3)
\]

where \(\rho\) is the mass density, and \(\sigma\) is the total stress tensor, which contains the electrostatic stress as well as the mechanical stress (Dorfmann and Ogden, 2005; Bustamante et al., 2009).

At the boundary between the body and the surrounding vacuum, the governing fields satisfy the jump conditions

\[
(\sigma - \sigma^*) \mathbf{n} = \mathbf{t}_m, \quad (\mathbf{d} - \mathbf{d}^*) \cdot \mathbf{n} = -\mathbf{w}_e, \quad (\mathbf{e} - \mathbf{e}^*) \times \mathbf{n} = \mathbf{0}, \quad (4)
\]

where \(\mathbf{t}_m\) is a prescribed mechanical traction and \(\mathbf{w}_e\) is the surface free charge density; herein and throughout the paper \((\cdot)^*\) denotes quantities in vacuum. Specifically, \(\sigma^*\) is

\[
\sigma^* = \varepsilon_0 \left[ \mathbf{e}^* \otimes \mathbf{e}^* - \frac{1}{2} (\mathbf{e}^* \cdot \mathbf{e}^*) \mathbf{I} \right], \quad (5)
\]

known as the Maxwell stress.

Assuming there is no interface free charge between adjacent phases \(m\) and \(f\), the jump conditions at internal boundaries are

\[
[\sigma] \mathbf{n} = \mathbf{0}, \quad [\mathbf{d}] \cdot \mathbf{n} = \mathbf{0}, \quad [\mathbf{e}] \times \mathbf{n} = \mathbf{0}, \quad (6)
\]

where \([x]\) is \((x)_{(m)} - (x)_{(f)}\); herein and henceforth the value of \((\cdot)\) in phase \(p\) is denoted by \((\cdot)^p\).

The total first Piola–Kirchhoff stress, Lagrangian electric displacement and Lagrangian electric field, respectively, are

\[
P = j \mathbf{F}^{-T}, \quad D = \mathbf{F}^{-1} \mathbf{d}, \quad \mathbf{E} = \mathbf{F}^{-1} \mathbf{e}; \quad (7)
\]

these satisfy the governing equations (3) in their Lagrangian form, namely

\[
\nabla_X \cdot \mathbf{P} = \rho \chi_{ttt}, \quad \nabla_X \cdot \mathbf{D} = 0, \quad \nabla_X \times \mathbf{E} = \mathbf{0}, \quad (8)
\]

where \(p_l = j \rho\). As discussed by Dorfmann and Ogden (2005), the transformation of the electric polarization density is not unique: it can be defined similarly to the transformation of the electric field or to the electric displacement. By choosing the analogue of the latter, the Lagrangian form of relation (2) reads

\[
P_e = \mathbf{D} - \varepsilon_0 \chi^{-1} \mathbf{E}, \quad (9)
\]

where \(\chi = F\mathbf{F}^{-T}\).

The fields \(\mathbf{P}\) and \(\mathbf{E}\) are derived from an augmented energy density function \(\Psi(F, D, X)\), as follows (Dorfmann and Ogden, 2005)

\[
\mathbf{P}_{\alpha} = \frac{\partial \Psi}{\partial F_{\alpha}} - p_0 \mathbf{F}^{-T}, \quad \mathbf{E} = \frac{\partial \Psi}{\partial D_{\alpha}}, \quad (10)
\]

where \(p_0\) is a Lagrangian multiplier accounting for the kinematic constraint \(f = 1\), if the material is incompressible; otherwise \(p_0\) is set to zero.

To analyze superposed incremental motions, we consider small time-dependent perturbations of \(\chi\) and \(D\), denoted by \(\chi(X,t)\) and \(D(X,t)\), respectively, accompanied by the increments \(\mathbf{p}\) and \(\mathbf{E}\) (Dorfmann and Ogden, 2010). Herein and in the sequel, increments are denoted by a superposed dot. Using \(x\) as a variable instead of \(X\), the governing equations of the incremental problem can be neatly written in terms of

\[
\Sigma = J^{-1} \mathbf{PF}^T, \quad \ddot{\mathbf{d}} = J^{-1} \mathbf{F} \mathbf{D}, \quad \ddot{\mathbf{e}} = \mathbf{F}^{-T} \mathbf{E}, \quad (11)
\]

which satisfy

\[
\nabla \cdot \Sigma = \rho \chi_{ttt}, \quad \nabla \cdot \ddot{\mathbf{d}} = 0, \quad \nabla \times \ddot{\mathbf{e}} = 0, \quad (12)
\]

where \(\chi(x,t) = \chi(X,t)\). The linearization of the constitutive relations of incompressible materials is

\[
\Sigma = \dot{\chi} \mathbf{h} + p_0 \dot{\mathbf{h}}^\mathbf{i} - p_0 \mathbf{l} + \mathbf{g} \ddot{\mathbf{d}}, \quad \ddot{\mathbf{e}} = \mathbf{B} \dot{\chi} + \mathbf{A} \ddot{\mathbf{d}}, \quad (13)
\]

where \(\mathbf{h} = \nabla \chi\) and on account of incompressibility

\[
\nabla \cdot \ddot{\mathbf{x}} = 0; \quad (14)
\]

herein \((\mathbf{B})_{\chi k l} = \mathbf{R}_{ijkl} h_{ij}\), and the components of the tensors \(\mathbf{A}\) and \(\mathbf{B}\) are

\[
\mathbf{A}_{ij} = j F_{\alpha}^{-1} \frac{\partial^2 \Psi}{\partial D_{\alpha} \partial D_{\beta}} J F_{\beta}^{-1}, \quad \mathbf{R}_{ijkl} = F_{\alpha \beta} \frac{\partial^2 \Psi}{\partial D_{\alpha} \partial D_{\beta}} J F_{\beta}^{-1}, \quad \mathbf{C}_{ijkl} = J^{-1} F_{\alpha \beta} \frac{\partial^2 \Psi}{\partial D_{\alpha} \partial D_{\beta}} J F_{\beta}^{-1}. \quad (15)
\]

\(^1\) For our proof of concept, we find this framework—which neglects viscosity—sufficient. A recent study on viscoelastic band gaps shows that accounting for viscosity does not have a fundamental effect on the band structure, but rather widens the gaps and shifts them towards lower frequencies (Zhu et al., 2016).
3. Motions of actuated soft dielectric fiber composite films

Consider a composite film whose phases are deformable dielectrics, and its thickness is $H$ in the $x_2$ direction. The film comprises fibers arranged periodically inside a matrix in the $(x_1, x_2)$ plane, such that the centers of adjacent fibers are at distance $A$ (Fig. 1(a)). The upper and lower surfaces are coated with stretchable electrodes, by which the film is actuated: when connected to a voltage source, the film deforms (Fig. 1(b)). The applied voltage is the only load—there are no mechanical tractions, and the film is free to expand in the $(x_1, x_2)$ plane. To model a general class of soft dielectric phases, we consider energy functions in the form

$$\Psi^{(\rho)} = W^{(\rho)}(I_1) + \frac{1}{2\varepsilon^{(\rho)}}\left(\mu^{(\rho)} I_{4e} + \gamma^{(\rho)} I_{5e} + \gamma_{F}^{(\rho)} I_{5e}\right),$$

(16)

where $I_1 = trC$, $I_{4e} = D : D$, $I_{5e} = D : C' D$, and $W^{(\rho)}(I_1)$ is any function of $I_1$; herein $\varepsilon^{(\rho)}$ agrees with the dielectric constant, such that $\varepsilon^{(p)} = \varepsilon^{(p)}I_0$, where $\varepsilon^{(p)}$ is the relative dielectric constant of phase $p$. Assuming the phases are perfectly bonded and their preferred direction is along the fibers, Getz et al. (2017) postulated a solution in the form of phase-wise constant fields, and showed that this solution satisfies required continuity and jump conditions. Furthermore, perfect bonding, in-plane symmetry and incompressibility further implied that the deformation is homogeneous, such that

$$\varepsilon^{(m)} = \varepsilon^{(f)} = \delta_{l,\lambda,\mu,\lambda},$$

(17)

The last of Eq. (6) showed that the electric field is homogeneous too

$$\varepsilon_{2f}^{(f)} = \varepsilon_{2m}^{(m)} = \varepsilon_2.$$

(18)

By satisfying Eq. (4) and the rest of Eq. (6), Getz et al. (2017) determined the relation between $\varepsilon_2$ and $\lambda$; for complete details we refer to Getz et al. (2017, Section 3). Subsequently, they have calculated the composite superposed motions, when approximated as a bulk. In what follows, we investigate incremental motions of the deformed film (Fig. 1(c)), while accounting for wave reflections from the top and bottom surfaces, and for the dependency on the $x_2$ coordinate, similarly to the model of Lamb (1917) in the limit of infinitesimal elasticity. Thereby, we extend the work of Getz et al. (2017), from bulk modes, to plate or Lamb modes of the actuated film.

We begin by specializing the equations governing linearized superposed motions to our settings. Accordingly, the first of Eq. (12) is

$$\rho(x)\ddot{x}_{1,tt} = \left[\left(\xi(x) + 2\mu(x)\right)\ddot{x}_{1,1} + \xi(x)\ddot{x}_{2,2} + \xi(x)\ddot{x}_{3,3}\right]_1$$

$$+ \left[\mu(x)\ddot{x}_{1,1} + \mu(x)\ddot{x}_{2,1} - \ddot{d}_{2}(x)\varphi_{1}\right]_2$$

$$+ \left[\mu(x)\ddot{x}_{1,1} + \mu(x)\ddot{x}_{3,1}\right] - \ddot{p}_{0,1},$$

(19)

$$\rho(x)\ddot{x}_{2,tt} = \left[\mu(x)\ddot{x}_{1,1} + \mu(x)\ddot{x}_{2,1} - \ddot{d}_{2}(x)\varphi_{1}\right]_1$$

$$+ \left[\xi(x)\ddot{x}_{1,1} + \xi(x)\ddot{x}_{2,2} + \xi(x)\ddot{x}_{3,3} - \ddot{d}_{2}(x)\varphi_{2}\right]_2$$

$$+ \left[\mu(x)\ddot{x}_{2,2} + \mu(x)\ddot{x}_{3,2} - \ddot{d}_{2}(x)\varphi_{3}\right] - \ddot{p}_{0,2}.$$  

(20)

The quantities $\mu(x), \xi(x), \varphi_{1}(x), \varphi_{2}(x), \varphi_{3}(x), d_{2}(x)$ are given in Appendix A; these quantities, as well as $\rho(x)$, are periodic in $x$, with the periodicity of deformed lattice. On account of last of Eq. (12), we write $\ddot{\varphi}(x, t) = -\ddot{\varphi}(x, t)$. Note that in contrast with the infinite case in Getz et al. (2017), the fields depend also on $x_2$. The set of equations is completed with the specialization of the incompressibility constraint (14) to our coordinates

$$\ddot{x}_{1,1} + \ddot{x}_{2,2} + \ddot{x}_{3,3} = 0,$$

(22)

and the incremental Gauss law

$$0 = \left[-\ddot{\varphi}(x, t)\ddot{x}_{1,1} - \ddot{d}_{2}(x)\ddot{x}_{2,1}\right]_1$$

$$+ \left[-\ddot{\varphi}(x, t)\ddot{x}_{2,2} - \ddot{d}_{2}(x)\ddot{x}_{2,2}\right]_2$$

$$+ \left[-\ddot{\varphi}(x, t)\ddot{x}_{3,3} - \ddot{d}_{2}(x)\ddot{x}_{3,3}\right]_3,$$

(23)

where $\ddot{\varphi}(x)$ and $\ddot{\varphi}(x)$ are given in Appendix A.

Following Vasseur et al. (2008) and Hou and Assouar (2008), in order to investigate wave propagation in a finite film we use the SC-PWE method. In this method, free boundary conditions at the top and bottom surfaces are implicitly accounted for using a fictitious periodicization of solid-vacuum layers, to enable a standard PWE also in the thickness direction (Laude et al., 2010). Accordingly, we employ a fictitious three-dimensional unit-cell by embedding the true two-dimensional unit-cell between two layers of vacuum (Fig. 2(a)). Hence, the fictitious super-cell has the dimensions $a \times l \times a$, where $a = \lambda A$, and $l = h + 2h_t$. Here, $h = \lambda^{-2}H$ is the thickness of the deformed film, and $h_t$ denotes the thickness of each vacuum layer in the fictitious unit-cell. The value $h_t$ is chosen according to numerical considerations, described later in Section 4. In electroelastic media, an implementation of the SC-PWE using the electric displacement field is accompanied with an erroneous
coupling of the electric fields between cells along the fictitious periodization. In this formulation, evanescent electric fields at different vacuum layers—related to the vacuum permittivity—incorrectly interact with each other. This interaction is not physical since the repetition of vacuum layers is artificial, in aim to set the stage for a solution using a PWE approach. For further details, the reader is referred to Laude et al. (2010, Section IV). A remedy was found by the latter group, using a formulation that is based on the electric polarization density instead on the electric displacement field. Thereby, since the polarization vanishes in vacuum, the unphysical interaction due to the fictitious periodization is avoided. Accordingly, we replace Eq. (23) with an equation of the electric polarization density (Appendix B), such that

$$0 = \left[-\varepsilon_0 \hat{\mathbf{E}}(\mathbf{x}) \cdot \hat{\mathbf{V}} \chi - \overline{\hat{\mathbf{p}}}_2(\mathbf{x})(\hat{x}_1 + \hat{x}_2)\right]_1 + \left[-\varepsilon_0 \hat{\mathbf{E}}(\mathbf{x}) \cdot \hat{\mathbf{V}} \chi - \overline{\hat{\mathbf{p}}}_2(\mathbf{x})(\hat{x}_2 + \hat{x}_3)\right]_2 + \left[-\varepsilon_0 \hat{\mathbf{E}}(\mathbf{x}) \cdot \hat{\mathbf{V}} \chi - \overline{\hat{\mathbf{p}}}_2(\mathbf{x})(\hat{x}_3 + \hat{x}_4)\right]_3,$$

where $\overline{\hat{\mathbf{p}}}_2(\mathbf{x})$, $\hat{\mathbf{E}}(\mathbf{x})$, $\hat{\mathbf{V}}$, and $\hat{\mathbf{E}}(\mathbf{x})$ are given in Appendix A. To validate the latter formulation, we have used it to evaluate—and indeed recover—band diagrams of the bulk modes examined in Shmuel (2013) and Getz et al. (2017).

To solve Eqs. (19)–(22) and (24) using the SC-PWE method, we expand $\rho$, $\hat{\mu}$, $\hat{\mu}_d$, $\hat{\mu}_\ell$, $\hat{\xi}$, $\hat{\xi}_d$, $\hat{\xi}_\ell$, and $\hat{\xi}_e$ to a threedimensional Fourier series. The expansion of each quantity, say $\zeta$, is

$$\zeta(\mathbf{x}) = \sum_{G} \zeta(G) \exp(iG \cdot \mathbf{x}),$$

$$\zeta(\mathbf{G}) = \frac{1}{v_G} \int_{v_G} \zeta(\mathbf{x}) \exp(-iG \cdot \mathbf{x}) d\mathbf{v},$$

where $\{\zeta(G)\}$ are the Fourier coefficients associated with the infinite set of reciprocal lattice vectors $G = \frac{2\pi}{a} n_1 \hat{\mathbf{i}} + \frac{2\pi}{b} n_2 \hat{\mathbf{j}} + \frac{2\pi}{c} n_3 \hat{\mathbf{k}}$, $n_1, n_2, n_3 \in \mathbb{N}$, and $v_G$ is the volume of the super-cell. Accounting for the super-cell periodicity, the Fourier coefficients become

$$\zeta(G) = \begin{cases} \left(\zeta^{(f)}(f) + \zeta^{(m)}(1 - \nu^{(f)}) \frac{h}{h'} + 2\zeta^{(s)}(s) \frac{h'}{h'} \right) \equiv \zeta, & G = 0, \\ \left(\zeta^{(f)} - \zeta^{(m)} \right) S_1(G) + 2(\zeta^{(s)} - \zeta^{(m)}) S_2(G) \equiv \Delta \zeta(G), & G \neq 0, \end{cases}$$

where

$$S_1(G) = \frac{1}{v_G} \int_{V_G} \exp(-iG \cdot \mathbf{x}) d\mathbf{v},$$

$$S_2(G) = \frac{1}{v_G} \int_{V_G} \exp(-iG \cdot \mathbf{x}) d\mathbf{v},$$

and $V_G$ and $V^*$ are the volumes captured by the fiber phase and each vacuum layer, respectively. Eqs. (27) and (28) are known as structure functions that depend on the shape of the fibers (see Kushwaha et al., 1994, Section IV). Next, using the Bloch theorem, we expand $\hat{x}(\mathbf{x}, t)$, $\hat{\phi}(\mathbf{x}, t)$ and $\hat{\rho}_0(\mathbf{x}, t)$ in the form

$$\hat{x}(\mathbf{x}, t) = \sum_{G} \hat{x}(G') \exp \left[i(G' + \mathbf{k}) \cdot \mathbf{x} - i\omega t \right],$$

$$\hat{\phi}(\mathbf{x}, t) = \sum_{G} \hat{\phi}(G') \exp \left[i(G' + \mathbf{k}) \cdot \mathbf{x} - i\omega t \right],$$

$$\hat{\rho}_0(\mathbf{x}, t) = \sum_{G} \hat{\rho}_0(G') \exp \left[i(G' + \mathbf{k}) \cdot \mathbf{x} - i\omega t \right],$$

where $\omega$ is the angular frequency and $\mathbf{k}$ is the three-dimensional Bloch wave vector. Here again, the summation in terms of $G'$ is carried out over the set of reciprocal lattice vectors, defined after Eq. (25). Expansions (25) and (29) are inserted into Eqs. (19)–(22), and after arithmetic manipulations, the result can be compactly written in the following matrix form (see Appendix C for the derivation):

$$Qu = \omega^{2} Ru,$$

where $u = \left[\hat{x}_1(G'), \hat{x}_2(G'), \hat{x}_3(G'), \hat{\phi}(G'), i\hat{\rho}_0(G')\right]^T$ contains the spectral components of the unknowns, and the components of the matrices $Q$ and $R$ are given in Appendix C. Eq. (30) defines a generalized eigenvalue problem for the eigenfrequencies $\omega$ at a given wave vector $\mathbf{k}$. In agreement with prior works (Vasseur et al., 2008; Brunet et al., 2008; El-Naggary et al., 2012), we have observed that computations with different values of $k_2$ yield similar results, as they should, since the periodicity in this direction is artificial. It implies that indeed there are no interaction between cells in the $x_2$ direction. Therefore, in the numerical investigation to follow, we set $k_2 = 0$, and consider Bloch vectors in the $(x_1, x_3)$ plane. We restrict attention to wave vectors at the boundary of the irreducible Brillouin zone, which defines the smallest region where wave propagation is unique (Kittel, 2005). For a square lattice, the zone boundary is defined by the lines connecting the points $\Gamma = (0, 0)$, $X = (\pi/a, 0)$, and $M = (\pi/a, \pi/a)$, as illustrated in Fig. 2(b); under certain conditions, scanning along this boundary is sufficient to find the extremum of the modes, and hence the gaps (Harrison et al., 2007; Craster et al., 2012).

A truncation of the infinite-dimensional system (30) is done by using a finite subset of $\{G\}$, i.e., a finite number of plane waves. We examined the relative change in band diagram frequencies using different number of plane waves (see, e.g., El-Naggary et al., 2012); in the subsequent parametric study, we found that the relative change of the band-gaps did not exceed 10% when increasing the number of plane waves from 1183 to 3087, corresponding to $-3 \leq n_2 \leq 3$ and $-6 \leq n_1, n_3 \leq 6$ or $-10 \leq n_1, n_3 \leq 10$, respectively.

### 4. Numerical investigation

To investigate the dependency of the motion on the properties of the phases, film thickness and in particular, bias electric load, we numerically evaluate the band structure in various cases. Choosing a study on films with circular fibers, the structure functions (27) and (28) obtain the form

$$S_1(G) = 2\nu^{(f)}f_1 \left(\frac{G_0}{G_2} \right) \frac{\sin \left(\frac{G_2}{2} \right) h}{G_2} \frac{1}{h'},$$

$$S_2(G) = \frac{\sin \left(\frac{G_2}{2} \right) \sin \left(\frac{G_2}{2} \right) \sin \left(\frac{G_2}{2} \right) h'/2}{G_2} \cos \left(\frac{G_2}{2} \right),$$

where $G_0 = \sqrt{4\pi^{(f)}(n_1^2 + n_2^2)}$ and $f_1$ is the Bessel function of the first kind of order 1. The constitutive behavior of each phase is described by the popular dielectric Gent model (e.g., Siboni and Castañeda, 2014; Zhao and Wang, 2014; Li et al., 2016).

$$\Psi^{(p)}(\mathbf{B}) = 2\mu^{(p)}f_1^{(p)} \frac{1}{2} \frac{1}{f_1^{(p)}} \left(1 - \frac{f_1^{(p)} h}{3} \right)$$

Herein, $\mu^{(p)}$ agrees with the shear modulus at small strains, and $f_1^{(p)}$ models the stiffening of elastomers near a limiting stretch.
We consider the following characteristic values of soft dielectrics (Kornbluh and Pelrine, 2008) for the matrix

\[ \rho^{(m)} = 1000 \text{ kg/m}^3, \quad \mu^{(m)} = 200 \text{ kPa}, \quad \epsilon_r^{(m)} = 3, \quad f_m^{(m)} = 10. \quad (34) \]

The chosen value of \( f_m \) corresponds to a limiting uniaxial stretch of 3.5. We examine different composites by varying the fiber properties in terms of the shear contrast parameter \( \alpha = \mu^{(f)}/\mu^{(m)} \), permittivity contrast parameter \( \beta = \epsilon_r^{(f)}/\epsilon_r^{(m)} \), and the locking contrast parameter \( \gamma = f_m^{(f)}/f_m^{(m)} \). Note that when \( \gamma < 1 \) (resp. \( \gamma > 1 \)) the fibers stiffen before (resp. after) the matrix as the deformation enhances. For simplicity, we fix \( \rho^{(f)} = \rho^{(m)} \).

In our numerical implementation, we used the following properties for the vacuum

\[ \rho^* = 0 \text{ kg/m}^3, \quad \mu^* = 0 \text{ kPa}, \quad \epsilon_r^* = 1. \quad (35) \]

We note that we did not encounter numerical instabilities associated with the huge mismatch in the properties of the vacuum and the solid, which is usually treated using the Lim method (Vasseur et al., 2008). Numerical instabilities appeared only when the thickness of the vacuum layers was set to be thicker than twice the deformed film thickness, i.e., \( h^* > 2\lambda^{-2}H \). This is in agreement with the report of Hou and Assouar (2008) and Laude et al. (2010), who did not observe instabilities when this criterion was satisfied. Therefore, we set \( h^* = 2\lambda^{-2}H \) in following numerical analysis.

To validate our SC-PWE implementation, we have evaluated the band structure of the examples in Getz et al. (2017), and recovered their results in the limit of a very thick film (\( \tilde{H} \approx 1000 \), where \( \tilde{H} = H/A \)).

We investigate first an exemplary composite with the characteristics

\[ \alpha = 10, \quad \beta = 10, \quad \gamma = 0.1, \quad \nu^{(f)} = 0.5, \quad \tilde{H} = 0.45. \quad (36) \]

Fig. 3 shows its normalized eigenfrequencies \( \tilde{\omega} = \omega A/2\pi c \), where \( c = \sqrt{\mu^{(m)}/\rho^{(m)}} \), as functions of the reduced wave vector \( \tilde{k} \) along \( \Gamma\Sigma\Gamma \), at different electric fields. Specifically, Fig. 3(a)-(c) correspond to \( \tilde{k} = 0, 1.8 \) and 3.2, respectively. The corresponding in-plane stretch ratios are \( \tilde{\lambda} = 1, 1.28 \) and 1.33, respectively. The gray and blue regions denote complete gaps and directional gaps, respectively; at frequencies of complete gaps, propagation is prohibited, independently of the propagation mode and direction; at frequencies of directional gaps, waves cannot propagate only along specific directions. As \( \tilde{k} \) enhances, we observe a shift towards higher frequencies for some modes along with a shift towards lower frequencies of others. This intricate manner of change causes an opening and closing of gaps as the electric bias field is increased. At \( \tilde{k} = 0 \), a complete gap covers the normalized frequencies \( 1.29 \lesssim \tilde{\omega} \lesssim 1.35 \); at \( \tilde{k} = 1.8 \) and \( \tilde{k} = 3.2 \) the complete gap is across \( 1.77 \lesssim \tilde{\omega} \lesssim 1.84 \) and \( 1.64 \lesssim \tilde{\omega} \lesssim 1.67 \), respectively. A directional gap appears in Fig. 3(b) in the \( \Gamma X \) section \( (x_1 \text{ and } x_3 \text{ directions}) \), across the normalized frequencies \( 1.33 \lesssim \tilde{\omega} \lesssim 1.52 \). We note that in Fig. 3(b) not all three fundamental modes, i.e., 0th antisymmetric \( \left( A_0 \right) \) and symmetric \( \left( S_0 \right) \) Lamb modes, and 0th transverse shear \( \left( T_0 \right) \), begin at the origin, as observed in prior works (Wilm et al., 2002; Hou and Assouar, 2008; Vasseur et al., 2008).

To identify the modes, we evaluate the displacements associated with each curve emanating from the origin along the thickness of the film, and examine their symmetry w.r.t. the mid-plane. Thus, Figs. 4 and 5 illustrate the displacements when \( \tilde{k} = 0 \) and 1.8, respectively, at \( \tilde{k} = 0 \), and arbitrary values of \( x_1, x_3 \) and \( t \). At \( \tilde{k} = 0 \), we identify the first, second and third modes, as the modes \( A_0, T_0 \) and \( S_0 \), respectively; at \( \tilde{k} = 1.8 \), we identify the first and second modes as the modes \( T_0 \) and \( S_0 \), respectively. Indeed, through a continuous evaluation of the band diagram as a function of \( \tilde{k} \), we observed that the mode \( A_0 \) is shifted towards lower frequencies until it vanishes. A similar shift and elimination of the mode \( A_0 \) in an actuated homogeneous film was observed by Shmuel et al. (2012).

The thickness of the film has a significant effect on the band diagram (see, e.g., Gao et al. 2007; Chen et al. 2006). We explore next the interplay between the initial thickness, the bias electric field, and the diagram. Specifically, Fig. 6(a)-(c) illustrate the prohibited normalized frequencies as functions of \( \tilde{k} = \tilde{H} = 0.45 \). 1 and 1.5, respectively, when \( \alpha = 10, \beta = 10, \gamma = 0.1 \) and \( \nu^{(f)} = 0.5 \). The complete gaps are denoted by the gray regions. When \( \tilde{H} = 0.45 \) and \( \tilde{k} = 0 \), a gap covers the frequencies \( 1.30 \lesssim \tilde{\omega} \lesssim 1.37 \); the application of voltage narrows the gap, up to its closure. Two gaps appear at \( \tilde{k} \approx 1.4 \) and \( \tilde{k} \approx 3.16 \), covering the frequencies \( 1.81 \lesssim \tilde{\omega} \lesssim 1.88 \) and \( 1.73 \lesssim \tilde{\omega} \lesssim 1.75 \), respectively, and close at \( \tilde{k} \approx 2.2 \) and \( \tilde{k} \approx 3.22 \), respectively. With \( \tilde{H} = 1 \), no gaps appear without a bias electric field. A gap opens when \( \tilde{k} \approx 1.3 \) at \( \tilde{\omega} \approx 1.34 \), and closes when \( \tilde{k} \approx 2.4 \) at \( \tilde{\omega} \approx 1.47 \). A maximal width of \( \Delta \tilde{\omega} \approx 0.05 \) is ob-
Fig. 4. The displacements $\dot{x}_1$ (black), $\dot{x}_2$ (red) and $\dot{x}_3$ (blue) along the thickness of the film for the modes (a) $A_0$, (b) $T_0$ and (c) $S_0$, in arbitrary units (a.u.). The displacements are evaluated at $k = \frac{\pi}{2} k_1 + \frac{\pi}{2} k_3$, and arbitrary values of $x_1$, $x_3$ and $t$, when $\hat{e} = 0$. The composite properties are $\rho^{(m)} = \rho^{(f)} = 1000 \text{ kg/m}^3$, $\mu^{(m)} = 200 \text{ kPa}$, $\epsilon_r^{(m)} = 3$, $f_0^{(m)} = 10$, $\alpha = 10$, $\beta = 10$, $\gamma = 0.1$, $\nu^{(f)} = 0.5$ and $\hat{H} = 0.45$. (For interpretation of the references to color in this figure legend, the reader is referred to the web version of this article.)

Fig. 5. The displacements $\dot{x}_1$ (black), $\dot{x}_2$ (red) and $\dot{x}_3$ (blue) along the thickness of the film for the modes (a) $T_0$ and (b) $S_0$, in arbitrary units (a.u.). The displacements are evaluated at $k = \frac{\pi}{2} k_1 + \frac{\pi}{2} k_3$, and arbitrary values of $x_1$, $x_3$ and $t$, when $\hat{e} = 1.8$. The composite properties are $\rho^{(m)} = \rho^{(f)} = 1000 \text{ kg/m}^3$, $\mu^{(m)} = 200 \text{ kPa}$, $\epsilon_r^{(m)} = 3$, $f_0^{(m)} = 10$, $\alpha = 10$, $\beta = 10$, $\gamma = 0.1$, $\nu^{(f)} = 0.5$ and $\hat{H} = 0.45$. (For interpretation of the references to color in this figure legend, the reader is referred to the web version of this article.)

Fig. 6. Prohibited normalized frequencies as functions of the electric bias field $\hat{e}$ at (a) $\hat{H} = 0.45$, (b) $\hat{H} = 1$ and (c) $\hat{H} = 1.5$. The composite properties are $\rho^{(m)} = \rho^{(f)} = 1000 \text{ kg/m}^3$, $\mu^{(m)} = 200 \text{ kPa}$, $\epsilon_r^{(m)} = 3$, $f_0^{(m)} = 10$, $\alpha = 10$, $\beta = 10$, $\gamma = 0.1$ and $\nu^{(f)} = 0.5$. 

\[ \begin{align*} 
\frac{\dot{x}_1}{x} &= \begin{cases} 
0.01 & \text{if} \quad x < h^* + \frac{h}{2} \\
0.05 & \text{if} \quad h^* + \frac{h}{2} < x < h^* + \frac{3h}{4} \\
0.01 & \text{if} \quad h^* + \frac{3h}{4} < x < h^* + h \\
0.0 & \text{otherwise} 
\end{cases} \\
\dot{x}_2 &= \dot{x}_3 \\
\frac{\dot{x}_1}{x} &= \begin{cases} 
-0.01 & \text{if} \quad x < h^* + \frac{h}{2} \\
0.05 & \text{if} \quad h^* + \frac{h}{2} < x < h^* + \frac{3h}{4} \\
0.01 & \text{if} \quad h^* + \frac{3h}{4} < x < h^* + h \\
0.0 & \text{otherwise} 
\end{cases} \\
\frac{\dot{x}_1}{x} &= \begin{cases} 
0.01 & \text{if} \quad x < h^* + \frac{h}{2} \\
0.05 & \text{if} \quad h^* + \frac{h}{2} < x < h^* + \frac{3h}{4} \\
0.01 & \text{if} \quad h^* + \frac{3h}{4} < x < h^* + h \\
0.0 & \text{otherwise} 
\end{cases} 
\end{align*} \]
tained when \( \dot{e} \simeq 1.8 \), at a central frequency of \( \dot{\omega} \simeq 1.40 \). The second gap opens when \( \dot{e} \simeq 2.2 \) at \( \dot{\omega} \simeq 1.73 \), and is shifted towards higher frequencies as \( \dot{e} \) increases. At \( H = 1.5 \), a higher electric field is needed to open a gap, namely, \( \dot{e} \simeq 2 \). Again the gap is shifted towards higher frequencies as \( \dot{e} \) is enhanced.

We investigate next the dependency of the band diagram on the physical properties of the constituents. Fig. 7(a) shows the complete gaps as functions of \( \alpha \) at \( \beta = 10 \), \( \gamma = 0.1 \) and \( \nu^f = 0.5 \) when \( \dot{e} = 0 \) (gray region), \( \dot{e} = 1.8 \) (blue region) and \( \dot{e} = 3.2 \) (red region). The minimal shear contrast observed for a gap opening is \( \alpha \simeq 6 \). The gap obtained for \( \dot{e} = 0 \) opens when \( \alpha \simeq 6 \) at \( \dot{\omega} \simeq 1.28 \), and becomes wider. The other two gaps appear only in certain ranges of \( \alpha \). Specifically, the gap obtained for \( \dot{e} = 1.8 \) opens when \( \alpha \simeq 7 \) at \( \dot{\omega} \simeq 1.82 \) and closes when \( \alpha \simeq 23 \) at \( \dot{\omega} \simeq 1.95 \). Its maximal width is observed when \( \alpha \simeq 13 \) being \( \Delta \dot{\omega} \simeq 0.09 \). at a central frequency of \( \dot{\omega} \simeq 1.88 \).

Fig. 7(b) displays the prohibited normalized frequencies as functions of \( \beta \) at \( \alpha = 10 \), \( \gamma = 0.1 \) and \( \nu^f = 0.5 \) when \( \dot{e} = 1.8 \) (blue region) and \( \dot{e} = 3.2 \) (red regions). The minimal permittivity contrast observed for a gap opening is \( \beta \simeq 2 \). For \( \dot{e} = 1.8 \), a gap opens at \( \beta \simeq 2 \), about \( \dot{\omega} \simeq 1.78 \); as \( \beta \) increases, it shifts towards lower frequencies until \( \beta \simeq 4 \), then shifts towards higher frequencies. A maximal width of \( \Delta \dot{\omega} \simeq 0.08 \) at a central frequency of \( \dot{\omega} \simeq 1.89 \) is observed when \( \beta \simeq 14 \), then it becomes narrower when the contrast is increased.

Fig. 7(c) illustrates the prohibited normalized frequencies as functions of \( \gamma \) at \( \alpha = 10 \), \( \beta = 10 \) and \( \nu^f = 0.5 \) when \( \dot{e} = 1.8 \) (blue region) and \( \dot{e} = 3.2 \) (red regions). No gaps were observed beyond \( \gamma \simeq 0.81 \). For \( \dot{e} = 3.2 \), a gap opens when \( \gamma \simeq 0.1 \) at \( \dot{\omega} \simeq 1.73 \) and repeatedly closes and reopens with the increase of \( \gamma \). The maximal width is observed when \( \gamma \simeq 0.36 \), and equals \( \Delta \dot{\omega} \simeq 0.18 \) at a central frequency of \( \dot{\omega} \simeq 1.73 \).

Fig. 7(d) shows the complete gaps as functions of \( \nu^f \) at \( \alpha = 10 \), \( \beta = 10 \) and \( \gamma = 0.1 \) when \( \dot{e} = 0 \) (gray region), \( \dot{e} = 1.8 \) (blue region) and \( \dot{e} = 3.2 \) (red region). The gaps do not appear at volume fractions lower than \( \nu^f \simeq 0.37 \) and greater than \( \nu^f \simeq 0.59 \). The gap

![Figure 7](image-url)
Table 1
Physical properties of silicone CF19-2186 and polyurethane PT6100S.

<table>
<thead>
<tr>
<th>Material</th>
<th>ρ (kg/m³)</th>
<th>μ (kPa)</th>
<th>εᵣ</th>
<th>f₀</th>
<th>Dielectric strength (MV/m)</th>
</tr>
</thead>
<tbody>
<tr>
<td>Silicone CF19-2186</td>
<td>1000</td>
<td>333</td>
<td>2.8</td>
<td>46.3</td>
<td>235</td>
</tr>
<tr>
<td>Polyurethane PT6100S</td>
<td>1200</td>
<td>5667</td>
<td>7</td>
<td>6.67</td>
<td>160</td>
</tr>
</tbody>
</table>

Fig. 8. Band diagrams of a composite made of silicone CF19-2186 matrix and circular polyurethane PT6100S fibers. The initial film thickness is $H = 1$ mm. The referential radius of the fibers is $R = 2.5$ mm, and the volume fraction is $\nu = 0.5$. The diagrams are evaluated at (a) $\varepsilon = 0$ MV/m, (b) $\varepsilon = 160$ MV/m, and (c) $\varepsilon = 235$ MV/m. The eigenfrequencies $\omega$ are shown as functions of the reduced wave vector $k$ along $\Gamma X M$. The gray region corresponds to a complete gap.

obtained for $\varepsilon = 0$ opens when $\nu = 0.37$ at $\nu = 1.19$, and closes at $\nu = 0.57$. A maximal width of $\Delta \nu = 0.065$ is obtained when $\nu = 0.5$, at a central frequency of $\nu = 1.34$. The gaps obtained for $\varepsilon = 1.8$ and 3.2 open when $\nu = 0.4$ at $\nu = 1.65$ and 1.5, respectively, and closes when $\nu = 0.59$ and 0.5, respectively; their maximal widths are $\Delta \nu = 0.08$ and 0.05, when $\nu = 0.5$ and 0.45, respectively.

We complete the numerical investigation with an analysis of a composite whose properties are of commercial products as in the example of Getz et al. (2017). Therein, the matrix and fibers are modeled as silicone CF19-2186 by Nusil and polyurethane PT6100S by Deerfield, respectively. We provide their relevant properties in Table 1. Upon setting the unit-cell dimensions, we note that current technologies enable 3D printing of features down to the microscale (Raney and Lewis, 2015). In what follows, we consider initial film thickness of $H = 1$ mm, with fibers of referential radius $R = 2.5$ mm, whose center-to-center distance is $A = 6.3$ mm, agreeing with the length-scale in Pelrine et al. (1998). Fig. 8 shows the dimensional eigenfrequencies $\omega$ as functions of the reduced wave vector $k$ along $\Gamma XM$, obtained at different values of electric field. Specifically, Fig. 8(a) correspond to $\varepsilon = 0$ MV/m, $\varepsilon = 160$ MV/m and $\varepsilon = 235$ MV/m, respectively. In the absence of electric field there are no gaps (Fig. 8(a)). At $\varepsilon = 160$ MV/m—the electric breakdown of polyurethane—there is an appreciable change in the diagram, although it is not accompanied with a gap opening. At $\varepsilon = 235$ MV/m, there is a complete gap across the frequencies 24.26 kHz $\leq \omega \leq 26.95$ kHz. While the complete gap appears at a voltage beyond the electric breakdown of the polyurethane, recent studies suggest that in the future, such electric loads will be accessible with the improvement of the dielectric strength of elastomers (Madsen et al., 2014; La and Lau, 2016). A continuous illustration of the change of the diagram as a function of $\varepsilon$ is shown in video 1 of the supplementary material online.

5. Summary

We have investigated the elastic band diagram of a film made of two soft dielectric phases, seeking to achieve tunable band gaps in practical structures. To this end, we have specialized the equations of small-amplitude waves in infinitely strained dielectrics, to the basic configuration of a film which was deformed by an applied voltage across its coating electrodes. The coefficients of the resultant coupled differential equations are position dependent, whose unknowns are functions of all three coordinates. To obtain a numerical solution, we have developed a variation of the SC-PWE method to our setting, which delivers a generalized matrix eigenproblem for the frequencies and modes. Using the method, we have executed a parametric study to explore how the band structure depends on the thickness of the film, the volume fraction of the constituents, and their physical properties. We found that for certain combinations of the above parameters, there exist directional and complete gaps, and that for other combinations there are no complete gaps, nor directional ones. Importantly, we found that by applying voltage (i) the location and width of existing gaps can be tuned; (ii) gaps can be opened in initially continuous diagrams. We emphasize that this electrostatically-controlled tuning strongly depends on the thickness of the film. Finally, we have examined the band diagram of a composite whose constituent properties correspond to commercial products. We have demonstrated that by applying voltage, a complete band gap is achieved; we note that the required electric field it significantly lower than the required in the bulk counterpart (Getz et al., 2017). We expect that our findings will promote realizing dielectric elastomer composites as active waveguides and isolators.

Acknowledgments

We thank Dr. Vincent Laude for helpful discussions. We also gratefully acknowledge the support of the Israel Science Founda-
tion, founded by the Israel Academy of Sciences and Humanities (Grant No. 1912/15) and the United States-Israel Binational Science Foundation support (Grant No. 2014358).

Appendix A

The quantities $\tilde{\mu}(x)$, $\tilde{\mu}(x)$, $\tilde{\zeta}(x)$, $\tilde{\xi}(x)$, $\tilde{d}_2(x)$, $\tilde{d}_2(x)$, $\tilde{e}(x)$, $\tilde{d}_2(x)$, $\tilde{p}_{c2}(x)$, $\tilde{p}_{c2}(x)$, $\tilde{\chi}_e(x)$ and $\tilde{\chi}_e(x)$ are

\[
\tilde{\mu}(x) = \tilde{\mu}(x)\lambda^2 , \quad \tilde{\zeta}(x) = \tilde{\zeta}(x)\lambda^4 , \quad \tilde{d}_2(x) = \tilde{d}_2(x)\lambda^{-2} , \quad \tilde{\zeta}(x) = \tilde{\zeta}(x)\lambda^{-8} , \quad \tilde{\xi}(x) = \tilde{\xi}(x) + \tilde{\mu}(x)(\lambda^{-4} + \lambda^2) , \quad \tilde{\chi}_e(x) = \tilde{\chi}_e(x) + \tilde{\mu}(x) \tilde{\chi}_e(x) + \tilde{\mu}(x)\lambda^{-4} .
\]

When specialized to Eq. (33), we have

\[
\tilde{\mu}(x) = \frac{\mu(x)}{1 - 2\lambda^2 + \lambda^2} .
\]

\[
\tilde{\zeta}(x) = \frac{\tilde{\zeta}(x)}{\tilde{\zeta}(x)} ,
\]

\[
\tilde{d}_2(x) = \tilde{d}_2(x)\lambda^{-2} , \quad \tilde{d}_2(x) = \tilde{d}_2(x)\lambda^{-8} , \quad \tilde{\zeta}(x) = \tilde{\zeta}(x) + \tilde{\mu}(x)(\lambda^{-4} + \lambda^2) , \quad \tilde{\xi}(x) = \tilde{\xi}(x) + \tilde{\mu}(x)\lambda^{-4} .
\]

Appendix B

We derive the linear approximation of the electric polarization density. This is done using an expansion of Eq. (9) as a Taylor series about the static current configuration, such that

\[
\tilde{\Phi}_e = \tilde{d} - \varepsilon_0 JF^{-T}E - \varepsilon_0 J\tilde{\Phi}_e - \varepsilon_0 J\tilde{\Phi}_e ,
\]

\[
\tilde{\Phi}_e = \tilde{d} - \varepsilon_0 JF^{-T}E - \varepsilon_0 J\tilde{\Phi}_e - \varepsilon_0 J\tilde{\Phi}_e ,
\]

where $J = JF^{-T}E$. Using Eq. (12), we can write $\tilde{\Phi}_e$ in terms of $\tilde{d}$, such that

\[
\tilde{\Phi}_e = (1 - \varepsilon_0 JF^{-T})\tilde{d} - \varepsilon_0 J\tilde{\Phi}_e - \varepsilon_0 (\tilde{h} + \tilde{h}^T - \tilde{\tau}h)e .
\]
When considering an incompressible homogenous material with an energy density function of the form
\[
\Psi = W_{\text{elas}}(l_1, l_2) + W_5(l_{5s}),
\]
the constitutive tensors \( \mathcal{A} \) and \( \mathcal{B} \) become
\[
\mathcal{A}_{ij} = \frac{1}{\bar{\varepsilon}} \delta_{ij},
\]
\[
\mathcal{B}_{ik} = \bar{\varepsilon} (\delta_{jk} d_i + \delta_{id} d_j),
\]
where \( \bar{\varepsilon} = (\frac{2 \mu_{0} \mu_{r}}{\mu})^{-1} \) and \( \varepsilon = \delta \mathbf{d} \). Note that \( W_{\text{elas}}(l_1, l_2) \) is a possibly nonlinear function of \( l_1 \) and \( l_2 = \frac{1}{2} [(tr C)^2 - tr \epsilon C^2] \), where \( W_5(l_{5s}) \) depends linearly on \( l_{5s} \). Applying Eqs. (A.5) and (A.6) to Eq. (A.3), while considering incompressibility, yields
\[
\mathbf{p}_s = \left[ 1 - \frac{\varepsilon_0}{\bar{\varepsilon}} \right] \bar{\varepsilon} \mathbf{d}.
\]
Thereby, the divergence of \( \mathbf{p}_s \) becomes
\[
\nabla \cdot \mathbf{p}_s = \left[ 1 - \frac{\varepsilon_0}{\bar{\varepsilon}} \right] \nabla \cdot \mathbf{d} = 0,
\]
where the last transition relies on Gauss law for \( \mathbf{d} \) given in the second of Eq. (12). The divergence of the electric polarization density equals to the minus of volumetric bound charge density. Hence, the incremental volumetric bound charge density vanishes in an incompressible homogenous dielectric media with an energy function of the form (A.4). Eq. (24) is obtained when specializing the constitutive law (16) to a case in which \( \gamma_0(-g) = \tau_2^{(g)} = 0 \) and \( \gamma_1^{(g)} = 1 \).

### Appendix C

Insertion of expansions (25) and (29) into Eqs. (19)–(22) provides
\[
\left\{ \sum_{G, G'} \left[ k_1(G') \left\{ \tilde{\mu}(G) \right\}(G' + k_1)(G + G' + k_1) + \frac{\mu(G)}{2} (G' + k_1) + \frac{\mu(G)}{2} (G + G' + k_1) + \frac{\mu(G)}{2} (G' + k_1) + \frac{\mu(G)}{2} (G + G' + k_1) \right] \right\} \exp(i(G' + \mathbf{x}) \cdot \mathbf{x}) \exp(i k \cdot \mathbf{x} - i \omega t) = 0,
\]
\[
\left\{ \sum_{G, G'} \left[ k_1(G') \left\{ \tilde{\mu}(G) \right\}(G' + k_1)(G + G' + k_1) + \frac{\mu(G)}{2} (G' + k_1) + \frac{\mu(G)}{2} (G + G' + k_1) + \frac{\mu(G)}{2} (G' + k_1) + \frac{\mu(G)}{2} (G + G' + k_1) \right] \right\} \exp(i(G' + \mathbf{x}) \cdot \mathbf{x}) \exp(i k \cdot \mathbf{x} - i \omega t) = 0.
\]

The sums in the curly brackets must vanish since the equations are valid for any \( \mathbf{x} \). We multiply these sums by exp \( i \mathbf{G}'' \cdot \mathbf{x} \), and integrate the product over the super-cell. It follows that in the summations over both \( G \) and \( G' \) only terms for which \( G'' = G' + G' \) remain, and in the summations over \( G' \) alone, only terms satisfying \( G'' = G' \) do not vanish. Thus, the final set of equations becomes
\[
\left\{ \sum_{G, G'} \left[ k_1(G') \left\{ \tilde{\mu}(G) \right\}(G' + k_1)(G + G' + k_1) + \frac{\mu(G)}{2} (G' + k_1) + \frac{\mu(G)}{2} (G + G' + k_1) + \frac{\mu(G)}{2} (G' + k_1) + \frac{\mu(G)}{2} (G + G' + k_1) \right] \right\} \exp(i(G' + \mathbf{x}) \cdot \mathbf{x}) \exp(i k \cdot \mathbf{x} - i \omega t) = 0.
\]

The sums in the curly brackets must vanish since the equations are valid for any \( \mathbf{x} \). We multiply these sums by exp \( i \mathbf{G}'' \cdot \mathbf{x} \), and integrate the product over the super-cell. It follows that in the summations over both \( G \) and \( G' \) only terms for which \( G'' = G' + G' \) remain, and in the summations over \( G' \) alone, only terms satisfying \( G'' = G' \) do not vanish. Thus, the final set of equations becomes
\[
\left\{ \sum_{G, G'} \left[ k_1(G') \left\{ \tilde{\mu}(G) \right\}(G' + k_1)(G + G' + k_1) + \frac{\mu(G)}{2} (G' + k_1) + \frac{\mu(G)}{2} (G + G' + k_1) + \frac{\mu(G)}{2} (G' + k_1) + \frac{\mu(G)}{2} (G + G' + k_1) \right] \right\} \exp(i(G' + \mathbf{x}) \cdot \mathbf{x}) \exp(i k \cdot \mathbf{x} - i \omega t) = 0.
\]
\[ Q_{G,G'}^{(3,3)} = \tilde{\mu}(G-G')(G'+k_1)(G+k_1) + \tilde{\mu}(G-G')(G'+k_2)(G+k_2) \\
+ \tilde{\mu}(G-G')(G'+k_2)(G+k_1) + 2\tilde{\mu}(G-G')(G'+k_2)(G+k_3) - \alpha^2 \rho(G-G') \]

\[ -\varphi(G')\tilde{d}_2(G-G')(G'+k_3)(G+k_2) + i\tilde{p}_0(G)(G+k_3) = 0, \quad (C.23) \]

\[ \chi_1(G)(G+k_1) + \chi_2(G)(G+k_2) + \chi_3(G)(G+k_3) = 0, \quad (C.24) \]

\[ \tilde{\chi}_1(G)(G+k_1) + \tilde{\chi}_2(G)(G+k_2) + \tilde{\chi}_3(G)(G+k_3) = 0, \quad (C.25) \]

\[ \chi_1(G)(G+k_1) + \chi_2(G)(G+k_2) + \chi_3(G)(G+k_3) = 0, \quad (C.26) \]

\[ \tilde{\chi}_1(G)(G+k_1) + \tilde{\chi}_2(G)(G+k_2) + \tilde{\chi}_3(G)(G+k_3) = 0. \quad (C.27) \]

\[ \chi_1(G)(G+k_1) + \chi_2(G)(G+k_2) + \chi_3(G)(G+k_3) = 0, \quad (C.28) \]

\[ \tilde{\chi}_1(G)(G+k_1) + \tilde{\chi}_2(G)(G+k_2) + \tilde{\chi}_3(G)(G+k_3) = 0. \quad (C.29) \]

\[ \chi_1(G)(G+k_1) + \chi_2(G)(G+k_2) + \chi_3(G)(G+k_3) = 0. \quad (C.30) \]

\[ \tilde{\chi}_1(G)(G+k_1) + \tilde{\chi}_2(G)(G+k_2) + \tilde{\chi}_3(G)(G+k_3) = 0. \quad (C.31) \]

\[ \chi_1(G)(G+k_1) + \chi_2(G)(G+k_2) + \chi_3(G)(G+k_3) = 0. \quad (C.32) \]

\[ \tilde{\chi}_1(G)(G+k_1) + \tilde{\chi}_2(G)(G+k_2) + \tilde{\chi}_3(G)(G+k_3) = 0. \quad (C.33) \]

\[ \tilde{\chi}_1(G)(G+k_1) + \tilde{\chi}_2(G)(G+k_2) + \tilde{\chi}_3(G)(G+k_3) = 0. \quad (C.34) \]

\[ R_{G,G'}^{(1,1)} = R_{G,G'}^{(2,2)} = R_{G,G'}^{(3,3)} = \rho(G-G'). \quad (C.35) \]

Blocks which are not listed are zero.

**Supplemental material**

Supplemental material associated with this article can be found, in the online version, at 10.1016/j.ijsolstr.2017.07.021.

**References**


